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Water to Zirconium and Zirconium Compounds

Editors: Barbara Elvers, Stephen Hawkins

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Contents

Water Waxes Weed Control Welding and C Whiskers Whitewares ... Wind Engrey Wine Wood Wood, Preserva

Wood, Surface Cross Referer

Weed Killers -+ Weld Cladiling Wheat → Cerea Whey -- Cheese Whisky - Spin Wollastonite --Wood Gasificati Wood Puip → P Xanthan Gum -Xanthene Dves Xenon → Noble oxide and hydrogen) is bubbled through. At reaction temperatures of 220-240 °C and pressures around 2 MPa, n-atkanes and n-atkenes are formed preferentially, with up to 40 % crude wax in the synthesis product.

Crude was is separated from the other products of synthesis by fractional condensation. The low-holling constituents (gasoline and diesel fuel) are removed by distillation under atmasphere pressure, followed by vacuum distillation to separate the soft maxes. Besides n-alkanes, disillation bottoms also contain alkeens, hydrocarbons with hydroxyl and carbonyl groups, and colored components. For purification and stabilization, the bottom products are subjected to a hydrofining step employing a nickel catalyst to yakla when hard was that is practically free of alkenes, aromatics, functional hydrocarbons, and milfur compounds.

Soft waxes are also subjected to hydrogenation.

Properties. Fischer 'Tropseh waxes consist constitution 'paparatifies with chain lengths between 20 and 59 carbon atoms. Products with an average molar mass of 490 g/mol are marketed as soft, and those with an average molar mass of 500 g/mol are marketed as soft, and those with an average molar mass of 500 g/mol as bard waxes. The paraffins have a fine crystalline structure and, because of the narrow molar mass distribution, a small melting range and very low melt viscosities. Congedaing point, density, and hardness incause with increasing mean molar mass. The low molar mass compared with polyolefin waxes is the reason for a cortain displacaciability of the crystall hypers relative to each other and, associated with this, the polisabality.

Synthetic paraflitis are fully compatible with refuned waxes, polyolefin waxes, and most vegetable waxes. They are solable at elevated temperature in the usual wax solvents (e.g., maphilat, turpentine, and tokene) to give dear solutions. Addition of 10 –20% synthetic parafit to other waxes increases their congealing point and hardness without significantly influencing metil viscosity, 10 away asies, the tendeevy of Fischer "Tropsels waxes to form microcrystals increases solvent retention. Some typical data for Poscher "Tropsels waxes are listed in Table 20.

Uses. Fischer—Tropach waxes are used in plastics processing as lubricants for polytyingl chloride) and polystyrene, as well as mold-release agents; as melting point improvers, hard-

Table 20. Typical properties of Pischer Tropsch with and hard paraffine

Characteristic	Sasol Wax M	Sasul Wax F12 Vestowax SP 1602		
Droo point, °C	~~~	106 - 112		
Needle penetration, 0.1 mm		13		
Color -	white	white		
Molar mass, u/mol	400	700		
Donsity (at 23 °C), p/cm ⁴	0.94	8.94		
Viscosity (120°C), mPa · s	< 20	< 20		

eners, and viscosity reducers in but melts and candles; and, because of their good polishability, for the production of cleaning agents and polishes. Micronized waxes improve the abrasion re-

sistance of paints and printing inks.

Oxidized waxes containing fatty acids and fall acid seters are produced from synthetic paratfin by oxidation and partial apportification. Major areas of application are as mode-release agents in plastics processing, in polishes and cleaning agents, and as auxiliaries in the textile and paper industries.

Trade Names. Fischer—Tropsch waxes are marketed, e.g., as Sasoi Wax (Sasoi Marketing Co., Johannesburg, South Africa) and Vestowax SH/SP (Hüls AG, Marl, Germany).

6. Polyolefin Waxes

6.1. Production and Properties

6.1.1. Polyethylene Waxes by High-Pressure Polymerization

High-pressure polychlylens (PE) waxes are produced, lich high-pressure polychlylene plasie, at high pressure and elevated temperature in the presence of radical formers. As waxes, their molar masses are considerably lower than those of plasties. The molar mass range is adjusted thiring polymerization by the addition of regulatures.

High-pressure polyethylene waxes are pattially crystalline and therefore consist mainly of branched molecular chains at which shorter side chains, such as ethyl and hutyl, predominate. They generally have polyethylene waxe branched polyethyle tallinity and density waxes, HDPE waxe creasing the pressur-

High-pressure pa lur masses betwee (weight-average moi gel permeation chr rently dominate the

The developmen proceeded in paralla (~Polyolefins, A2) waxike ethylene pol a variant of the IC tion process [6,3]. In pressure ethylene wa [6,2], [6,3]. Parallel thermal depolymeridation of polyethylawaxes. was develop

With the rapid go the expansion of pe the economic imporcreased rapidly. War erry profiles could be density and molar i and functionalizing



Figure 18. Schematic of 8) Precomprestor: b) P They generally have low densities (low density polyethylene waxes, LDPE waxes). Less branched polyethylene waxes with higher crystillinity and density chigh-density polyethylene waxes. HDPE waxes, and no produced by in-

creasing the pressure. High-pressure polyethylene waxes with mofern masses between 3000 and 20000 g/mol |weight-average moles mass $\tilde{M}_{\rm w}$, determined by gel permeation chromatography (GPC)| currently dominate the market.

The development of high-pressure PE wases proceeded in pariella to hist of LDPB plastics (~Polystlefins, A21, pp. 488–489). In 1939, wastike ethylene polymers were formed by using a variant of the ICI high-gressure polymerization process [6.1]. Industrial production of ligh-pressure ethylene warses was started in the 1940 [6.2], [6.3]. Parallel to this, the technology of thermal depolymentization—the thermal degradation of polyethylene plastics to polyethylene warse—was developed (see Section 6.1.4).

With the rapid growth of pet ochemistry and the expansion of polyethylene plant capacities, the economic importance of PE waxes also increased rapidly. Waxes with very versatile property profiles could be synthesized by varying the density and nodar miss of the homopolymers and functionalizing polyethylene by copolysmer-

ization with various monomers (e.g., vinyl acetate or acrylic acid) or by molt oxidation.

Because PE waxes are rather inexpensive, have improved applicability and consistency in quality, and are in constant supply they have displaced expensive natural waxes (e.g., carnaúba and montan waxes) in many areas.

The high-pressure process is not applicable for production of polypropylene waxes, because only soft or oily products can be obtained. These waxes are therefore produced by the Ziegler process (see Section 6.1.3) or by depolymerization of high-density polypropylene (see Section 6.1.4).

6.1.1.1. Production

The technology involved in the production of the production of the production of the production of the product of the product final processure polyethylene (—Polyoleffins, A21, pp. 506–509). The only differences are in the product finishing process. Unlike viscous plastic melts, which must be granulated underwater in granulators, for example, the mobile wax melt can be converted into powder by spraying or into granules by using dicers

The reaction vessels are stirred autoclaves or tubular reactors (~ Polyolefins, A21, pp. 507-508), the design of which can vary considerably.

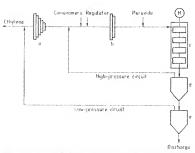


Figure 16. Schematic of production of polyethylene waxes by the high-pressure process in stirred autoclaves.

3) Precompressor; b) Posterappressor; c) Autoclave reactor; d) High-pressure separator; e) Low-pressure separator.

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Polymerization is terminated by recombination or disproportionation of two macroradi-

$$2R - CH_y - CH$$

Long-chain branching occurs by intermolecular chain-transfer reactions:

A particular feature of polyethylene wax synthesis by the high-pressure process is the increased formation of short ethyl and buryl side chains. These are formed by intramolecular rathcal transfer:

At the high pressures necessary for polymerization, othylene is in a supercritical state. Polymerization techniques are supercritical state. Polymerization therefore isless place in a one-phase system. After leaving the reactor, the reaction mixture is decompressed in high- or low-pressure separators, and unrecarded ethylene evaporates. The wax remains as a melt and ethylene is received.

Reaction Conditions. The structure, modurmass, and thus properties of polyethylene was are determined mainly by reaction pressure, reaction temperature, type and quantity of infitator and moder mass regulator, and reactor type and geometry [6.5]—[6.9].

In the homo- and copolymerization of ethylene the reaction pressure is usually 1520 MPn. Somewhat lower pressures are used in autoclaves than in tubular reaction. Other process variants involve much lower pressures of 70 MPn (max.) with isopropanol as the molar mass regulator [6,7], [6,8].

Higher pressure favors chain propagation and thus leads to very short residence times in the reactor. It also inhibits chain-tunsfer reactions to that with increasing pressure the degree of branching decreases and the density, crystallianly, luardinoss, and mp increase. In tubular reactors, rising pressure increases the polymer yield, whereas in autoclaves this effect is smaller.

For homopolymerization the reaction temperature is 200–350°C and for copolymerization temperature is 200–350°C. High reaction temperature favor chain-stransfer reactions. For this reason the polyhelynen waxes formed turn earny short-chain and only a few long-chain branches [6,10] and thus low densities. Higher-density waxes with lower degrees of branching are formed at lower reaction temperature.

Organic peroxides and molecular oxygen (the latter exclusively in tubular reactors) are used us initiators. They decompose into radicals under polymerization conditions. To ensure that the peroxides are metered reproducibly, they are dissolved in organic solvents.

solved in organic solvents. Hydrogen and almost all organic compounds can act as molar mast regulators (chain terminators). Hydrogen, lower alkams (e.g., propane), lower alkens (e.g., propine) alkens (e.g., propine) alkens (e.g., propine), lower alkens (e.g., propine) and investable of the solvent alkens (e.g., propine) and instruss of these authorities (e.g., propine) and instruss of these substances are undirty employed. The concentration and reactivity of these regulators determine the average under mass, the degree of branching, and thus the density of polyethylene wax. The activity of the regulator increases with increasing temperature and decreases with increasing pressure (e.4).

Reactor type and geometry also significantly influence polymer structure and properties. Because of the backmaxing occurring in antoclaves, spherical molecules with many long-chain branches are predominantly formed. In tubular reac-

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mpounds ferminatopane), et, alkyl iounideauoti or uly emof these mass, the nsity of egulator d decre-

ificantly ties, Benelares, in brunur reactors with plug flow, long, straight molecules with little long-chain branching predominate [6,13]. High-density, highly crystalline polyethylene wares can therefore be produced only in tubular reactors.

finat as the geometries of the autoclave and the tubular reactor differ, so to operating procedures. In the case of antoclaves the initiator solution and the melan mass regulator are charged directly to the preheated pressurized vessel by using high-pressure pumps. In tubular near-tors the initiator (e.g., air) and the molar mass regulator are added to the reaction mixture believe the compression stage or at the entraince to the reactor. With autoclaves, commonmers can be charged to different parts of the reaction vessel (see Fig. 10).

Product Pinishing, Shaping of the molten polyethylenc can be performed directly after ramoval of gaseous products in high- and low-pressure separators and subsequent fine degassing and filtration. Since no metal-containing cutalysts are used in high-pressure polymerization of ethylenc, no citalyst enrowed in secessary. Low medar mass fragments and residues of the initiation and medar mass regulator evaporate mainly in the separators. Only an extremely andl proportions is incorporated into the polysmall proportion is incorporated into the polysmall proportion.

Grandes can be produced by using strip dicars. The wax melt flows in thin strips on a continuous, water-cooled metal band and, after solidification, is chopped with a rotating knife. Pastillex can be obtained by metered dripping of the melt from a distributor head onto this type of cooling band.

A particular characteristic of wax melts in their low melt viscosity. Thus, apraying the nest to form postders of spherical particles is also possible. By naing special nozele arrangements (binary nazales) with nitrogen as the atomizing gas, micronized waxer (very fine powders) with particle sizes in the micrometer range can be produced.

Powders and micromized waxes can also be obtained by grunding granules in jet mills. With soft or very viscoelastic waxes, cooling with dry ice or liquid nitrogen is necessary. Coarse particles must be removed by subsequent classification.

Like all organic dusts (e.g., coal or flour), wax powders, and particularly micronized waacs, are highly susceptible to dust explosions. Appropriate safety precautions must be taken during processing, in particular grounding all installations to avoid spark formation through electrostatic charge buildup.

6.1.1.2. Properties

The transition from polyethylene plastic to polyethylene wax is flexible. If the molar mass is lowered, the thermoplastic materials is gradually changed to a wax. The DGF definition of wax (see Section 1.2) gives an approximate boundary: For waxes, an upper limit to the melt viscosity of ∞ . 20 000 mm//s at 1.20 °C is defined, which corresponds to an average molar mass M_w) of ca. 37000 m0 f. ca.

The properties of polystbylen waxes are determined strongly by \hat{M}_s as a measure of the average chain length and by the degree of burnching as a measure of the shape of the molecules. Mell viscosity increases with increasing molar mass. Crystallinity, hardness, m_p and solidification point increase as the degree of branching decreases. These data are important for application-oriented properties.

The molar mass can be determined by OPC. This method gives the weight-average (M_s) and the number-average (M_s) molar mass. The ratio M_sM_s is a measure of meltur mass distribution and is known as the polydispersity index (M_s) measurement of the polydispersity in the polydisper

Table 21 gives a series of structure and proporty data for three high-pressure polyethylene waxes with increasing density from two different producers (some of this represents unpublished data supplied by the authors).

The following conclusions can be drawn from these data:

- 1) The degree of branching is generally low with ca. ten branches (max.) per molecule. The side chanas are mostly chyd and butyl groups with a very small proportion of long-chain branches [6,12] - [6,14]
- The density, crystallinity, hardness, solidification point, and drop point increase as degree of branching decreases [6.15]. All densi-

hot melts.

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Characteristic	A	8	€.	D	9.	Į-
Density (23°C), g/cm² (DGP-M-IH 2a)	0.02	0.92	0.945	0.92	0.92	6.94
CH, per 1600 C alones	76	22	17	32	23	20
Branithos per moleculo	3.4	1.8	1.0	3.7	1.9	1.8
Crystallinity (from IR data), %	54	59	67	52	62	64
Central double bonds per 1000 C atoms	0.1	0.4	0.4	0	0	- O
Terminal double bonds per 1699 C atomis	0.1	0.1	9	1.3	1.2	1.3
Side-chain double bonds per 1000 C stoms	0.6	0.4	0.2	6.9	0.5	0.4
Bell indentation hardness (23°C), but	200	370	550	314	413	482
(DGF-M-III 9a)						
Solidification point, *C (DGF-M-III 4a)	96	99	106	43	801	104
Drog point, °C (DCH-M-IR 3)	106	112	116	106	133	113
Melt viscosity (176 °C), mm2/s (DOF-M-IH 8)	1310	1160	1200	780	925	890
Weight-average moist mass M., g/moi	6140	5930	6000	5300	5500	5500
Number average moder tokes M., g/mod	2920	2396	2440	2490	2380	2566
M_M, (polydispersity index)	2.1	2.5	2.5	2.1	2.3	2.1

and a second second

Characteristic	Ethylene-vinyl scelate copolymer wax	Ethylens -acrylic non copolymer wax
Weight-average molar mass M _w , g/mol	ca. 6800	cs. 6100
Number average motar mass M., gimot	3000	3000
Number average mour mass m_{μ} , grand Melt viscosity (120°C), mur'/s (DGF-M-HI S)	2000	£300
Solidification point, *C (DGF-M-Hi %)	85	93
Drop point, C (DGF-M-III 3)	96	102
Ball indentation hardness (120°C), but (DGF-M-I/I 9a)	120	410
Acid number, me KOH/g	120	45
Acid number, ing & Odig Vinyi goztaje content. %	iti	

ty-dependent properties vary correspondingly.

3) High-pressure PE waxes contain a small proportion of double bonds, whose distribution on the macromolecule depends on the production process, as the large differences between the levels of central double bonds show. The formation of double bonds is attributed to depolymerization of chain radicals at high reaction temperature [6, 4], [6, 16], e.g.

$$R_1$$
 CH · QH · R_2 · · · · · · R_3 - CH · CH · R_3 + \hat{R}_3

High-density polyethylene waxes me colorless, white to transparent and form clear mels; Like other waxes, they dissolve in compolar solvents (e.g., aliphatic, aromatic, and ethorismal hydrocarbons) on beaung and generally crystallize as very fine particles on cooling. Depending on the type and concentration of the wax, they then form mubile dispersions or poste-like gets, which frequently exhibit historropic properties.

6.1.2. Copolymeric Polyethylene Waxes by High-Pressure Polymerization (→ Polyolefins, A21, pp. 508-509)

In the high-yessure process, many other monomers can copolymeria with ethylene give monomers can copolymeria with ethylene give ing rise to considerable changes in product properties [6-7], [6-9]. For copolymerizano, autocalews with their intensive mixing, constant temperature, and consequently stable reaction process are particularly suitable. In industry, vinyl acetate and acrylic acid are used mainly as comonomers, giving waxes with higher polarity and lower crystallinity. Here also, the degree of branching decreases with increasing pressure and decreasing temperature. Density, mp., and hardness increase accordingly.

Table 22 gives characteristic data for two important copolymeric high-pressure polyethyl-ene waxes with vinyl acetate and acrylic acid as commonomers.

Ethylene-vinyl acetate copolymer is a relatively polar but nevertheless hydrophobic wax that can be dispersed in organic solvents particularly well it is a automotive pain pigment concents

Ethylene---ner, for aqueous syste reachly and is ch sive. Its enrulsion floor polishes and

6.1.3. Polyclella Polymerization

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PE waxes can be
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propylene or high Natta process is pp. 522-529). Ziegler PE w

Zieger Pf. w have a rooter ma-(M_a). They thus b Tropach and para plastics. Besides polymerization p degree of crystall high-melting proucts with low meucts with low me-

The soluble, produced in the are more similar waxes because o normal hard wax only after process [6,17].

6.1.3.1. Producti-

For the pred olefin waxes, Ziej with low overag chain-length dist active under the thesis, are parties of polymerization carrying out pol 483

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Vol. A 28 factly well. It is used as an additive in metallic automotive paints, as a dispersing agent in 0.94 20 1.8 64 0 pigment concentrates, and as a component of but melts Ethylene - acrylic acid copolymer is suitable 6.4

for aqueous systems because it can be emulsified readily and is chemically and thermally intensive. Its emulsions are used mainly in modern floor polishes and mold-release agents.

6.1.3. Polyolefin Waxes by Ziegler-Natta **Polymerization**

Like high molar mass polyethylene plastics. PE waxes can be produced by the Ziegler lowpressure process using organometallic catalysts (-+ Polyolefins, A21, pp. 509-511), in addition to the radical process at high pressure and temperature. Like high-density polyethylene, Ziegler waxes have a mainly linear molecular structure, They can contain short side chains but not long ones like high-pressure polyethylene waxes.

For direct synthesis of polymer waxes from propylene or higher x-oletins, only the Ziesler ... Natta process is suitable (→ Polyolefins, A21, pp. 522 - 529).

Ziegler PE waxes currently on the market have a molar mass between 800 and 8000 g/mol (Ma). They thus bridge the gap between Pischer-Tropsch and paraffin waxes and HDPE thermoplastics. Besides controlling chain length, the polymerization process allows adjustment of the degree of crystallinity so that both hard - brittle, high-melting products and softer, flexible products with low melting points can be obtained.

The soluble, soft wax fractions inevitably produced in the suspension process for HDPE are more similar to paratifu waxes than to PE waxes because of their low molar masses. Por normal hard wax applications, they can be used only after processing by distillation or extraction

6.1.3.1. Production

For the production of Ziegler Natta polyolefin waxes, Ziegler catalysis that give polymers with low average chain lengths and narrow chain-length distributions, and are sufficiently active under the special conditions of wax synthesis, are particularly suitable. The low degree of polymerization typical of waxes is achieved by carrying out polymerization in the presence of

hydrogen as a molar mass regulator at comparatively high temperature (usually between 100) and 200°C, sometimes even higher) [6.18] -[6.22] Aliphatic hydrocarbons are used as reaction medium. Because of the high reaction temperature the waxes are formed in solution (solution polymerization). If the product viscosity is sufficiently low the molten polymer tormed can itself function as the solvent (bulk polymerization) [6.23], [6.24]. Unlike modern HDPE processes in which the catalyst is not removed, in the case of wax synthesis the catalyst must usually be decomposed and illiered off [6.25], [6.26]. This applies at least to the heterogeneous maniana catalysts still used today (see below) and is necessary because waxes have higher purity requirements than PE plastics as a consequence of the applications for which they are used. After the solvent has been distilled off, the wax melt is shaped (e.g., by spraying or pastille formation).

Catalyst Systems. Classical Ziegler catalysts, consisting of titanism terrachloride and alkylaluminum compounds, can be used for wax synthesis, but their activity is low [6.23], [6.24], [6.27]-[6.29]. The degree of branching and thus also the density, drop point, and hardness can be adjusted by copolymerization, by the special type of catalyst preparation [6.28], and by varying the polymerization temperature [6 23], [6,29].

The state of the art for production of PE homo- and copolymer waxes involves, as in piastics production, the use of supported catalysts, which contain titanium atems as the active species and magnesium compounds as the carrier material (-+ Polyolefins, A21, p. 593). Catalysts derived from titanium tetrachloride and magnesinm chloride, oxide, hydroxide [6.19] [6.22], or alkoxide [6.18], for example, are suitable.

Catalysis based on Ti-Mg compounds can also be used in the production of polypropylene waxes. Through an appropriate choice of catalyst, the degree of crystallimity (isotacticity) (→ Polyolefins, A21, pp. 532 - 534) can be varied within wide limits and thus adapted to a particu-Lir application. Flexible products with average crystallinity can be obtained [6,30]. Highly crystalline polypropylene waxes can now be produced economically by using stereoregulating silanes as additional catalyst components [6,31], [6.32]. Pormerly, these waxes could be produced only by thermal degradation of high molar mass polypropylene (Seenon 6.1.4)

WO 00/50473 PCT/U800/04754

The ethylene acid-containing copolymer aqueous dispersion of the present invention consists essentially of a dispersion of component (A), an ethylene-methacrylic acid copolymer containing 15-35 wt% of methacrylic acid as the ethylene acid-containing copolymer, in water in the presence of component (B), animonia used as a basic component in an amount greater than the amount of the carboxyl groups of component (A).

Both good dispersion property and good dispersion stability can be obtained by using an excess of component (B) animonia, particularly an amount sufficient for neutralizing 110-150% of the carboxyl groups of the abovementioned acid-containing copolymer (A). The resulting aqueous dispersion can be coated onto a substrate, such as a film, to make a coated substrate, particularly a coated film, that is not susceptible to moisture accumulation and has a good waterproofness.

It is suitable for ethylene-methacrylic acid copolymer (A) to contain 15-35 wt% or alternatively 15-25 wt%, particularly 18-30 wt%, of an unsaturated carboxylic acid. In the case of using a copolymer containing an unsaturated carboxylic acid in an amount that is less than the above-mentioned range, it is difficult to obtain a composition having a good aqueous dispersion property. In the case of using a copolymer containing an unsaturated carboxylic acid in an amount that is more than the above-mentioned range, a stable dispersion composition cannot be obtained and both the waterproofness and mechanical strength of the coated film are reduced.

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An ethylene-methacrylic acid copolymer having a melt flow rate of 50-2000 grams/10 minutes, particularly 60-1500, at 190°C/2160 gram load is suitable. In the case of using a methacrylic acid copolymer having an extremely low melt flow rate, an aqueous dispersion composition having a good dispersion property cannot be obtained. When using a copolymer having an excessively high melt flow rate, the coated film has a poor strength.

Besides ethylene and methacrylic acid, the copolymer may be copolymerized with other monomers including an unsaturated carboxylic acid ester such as methyl acrylate, ethyl acrylate, n-butyl acrylate, isobutyl acrylate, methyl methacrylate, isobutyl methacrylate, dimethyl maleate, or diethyl maleate; a vinyl ester such as vinyl acetate or vinyl propionate; and carbon monoxide, in an amount of 20 wt% or less, particularly 10 wt% or less.

The aqueous dispersion composition of the present invention

40 contains anmonia, which can neutralize an excess, particularly 110-150%, more
particularly 120-140% of the carboxyl groups of copolymer (A), along with
copolymer (A).